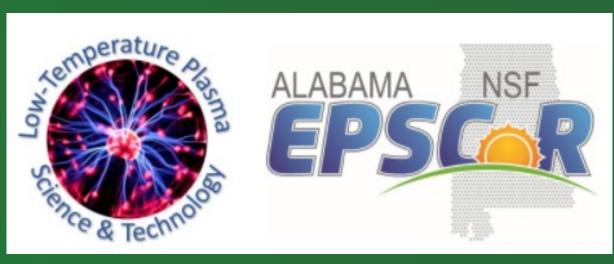
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MoNbTaVWC<sub>5</sub>, EFA=117 (eV/atom)<sup>-</sup>

### Aims:

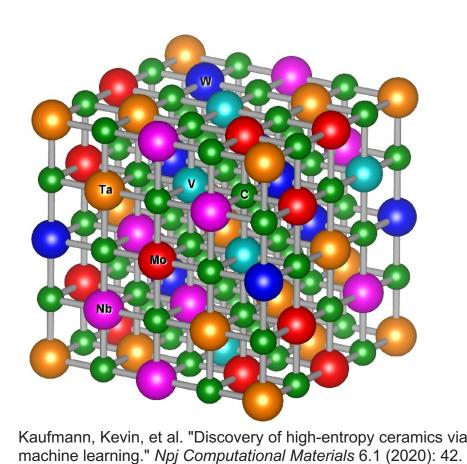
- Explore the novel approach of using microwave induced plasma to synthesize high entropy carbides
- Evaluate the propensity to form a solid solution rock-salt structure based on the high entropy carbide:  $(Hf_{0.2} Zr_{0.2} Ti_{0.2} Ta_{0.2} Nb_{0.2})C$

## Microwave Plasma Annealing:

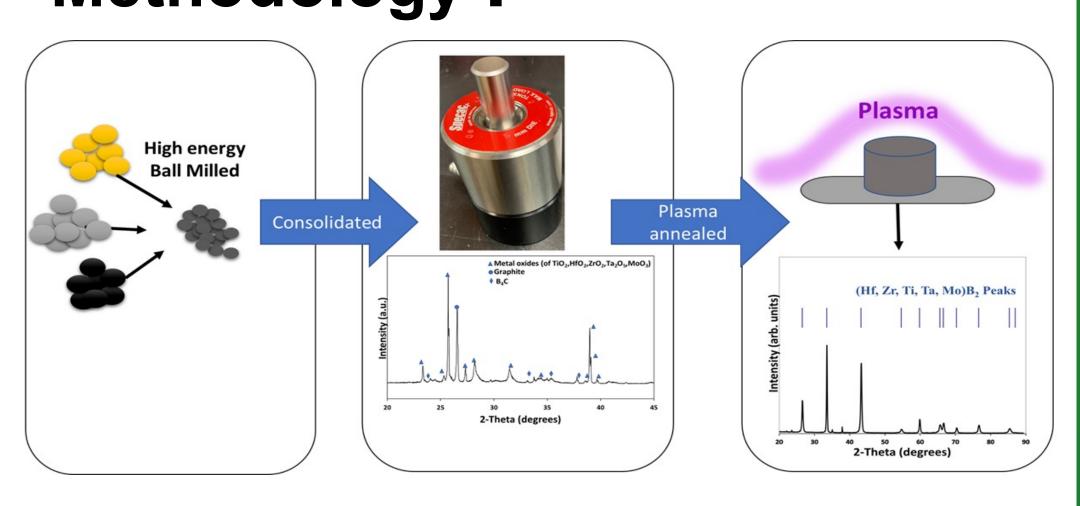
- Rapid MW heating of ceramic core
- Plasma discharge promotes chemical reactions as highly active species
- Potential benefits: Reduced sintering temperature, increased densification rate, and minimized grain coarsening

## High Entropy Carbides:

- Rock-salt crystal structure
- Known for thermal stability, high strength and modulus, good oxidation resistance and can withstand extreme conditions



# Methodology:

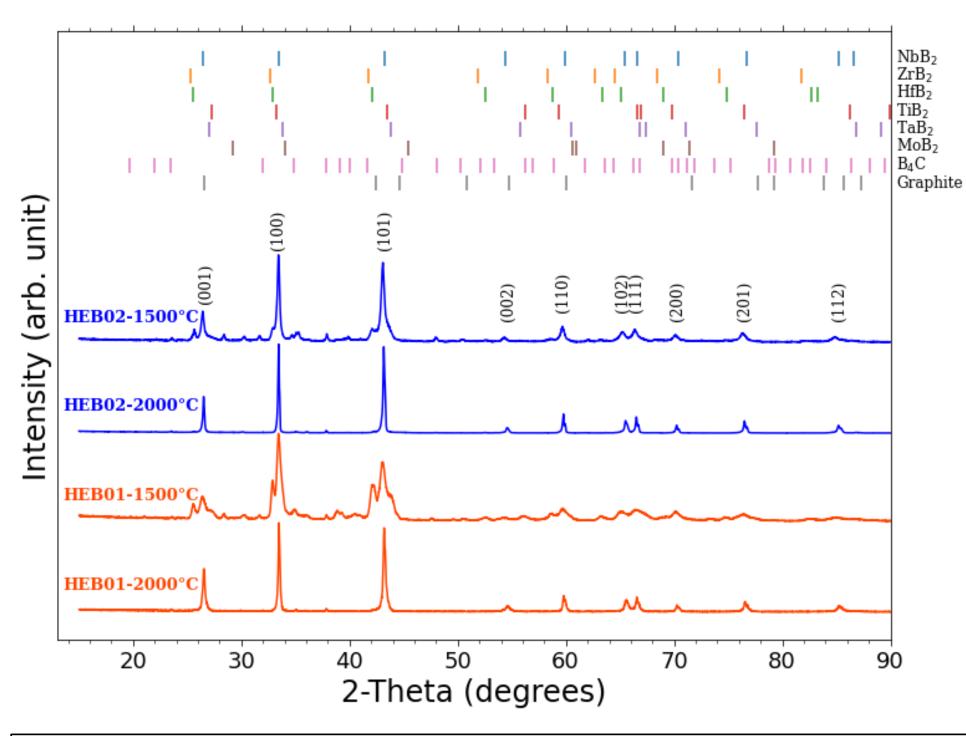


Start with metal oxides + graphite as reducing agent

High energy ball mill then press into pellet.

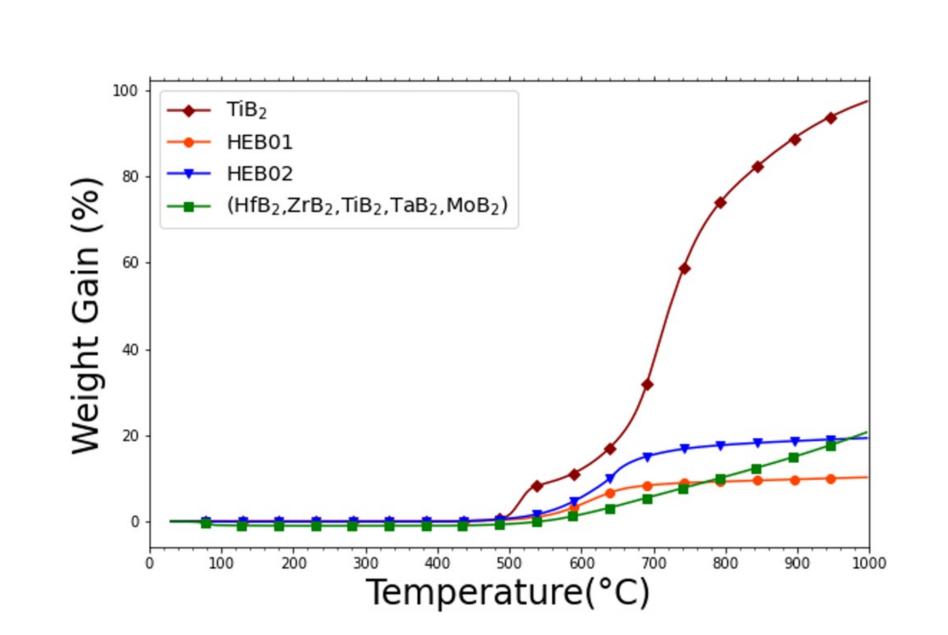
The pellet is MW plasma annealed to 1800°C or 2000°C for 1hr in an argon-rich

# Prior Success in High-Entropy Borides:



XRD for two HEB configurations plasma annealed at 1500 °C and 2000 °C. The Entropy Forming Ability (EFA) is higher for HEB02, resulting in more complete conversion to the high entropy phase at lower annealing temperature.

 $\rightarrow$  HEB01: HfZrTiTaMoB<sub>10</sub> EFA: 207(eV/atom)-1  $\rightarrow$  HEB02: HfZrNbTaMoB<sub>10</sub> EFA : 294(eV/atom)-1



Weight gain (in air) vs. sample temperature measured from thermogravimetric analysis. Samples measured include the plasma annealed HEBs made at 2000 °C: HEB01 (orange), HEB02 (blue), a mixture consisting of five commercial diborides (green), and TiB2 (brown).

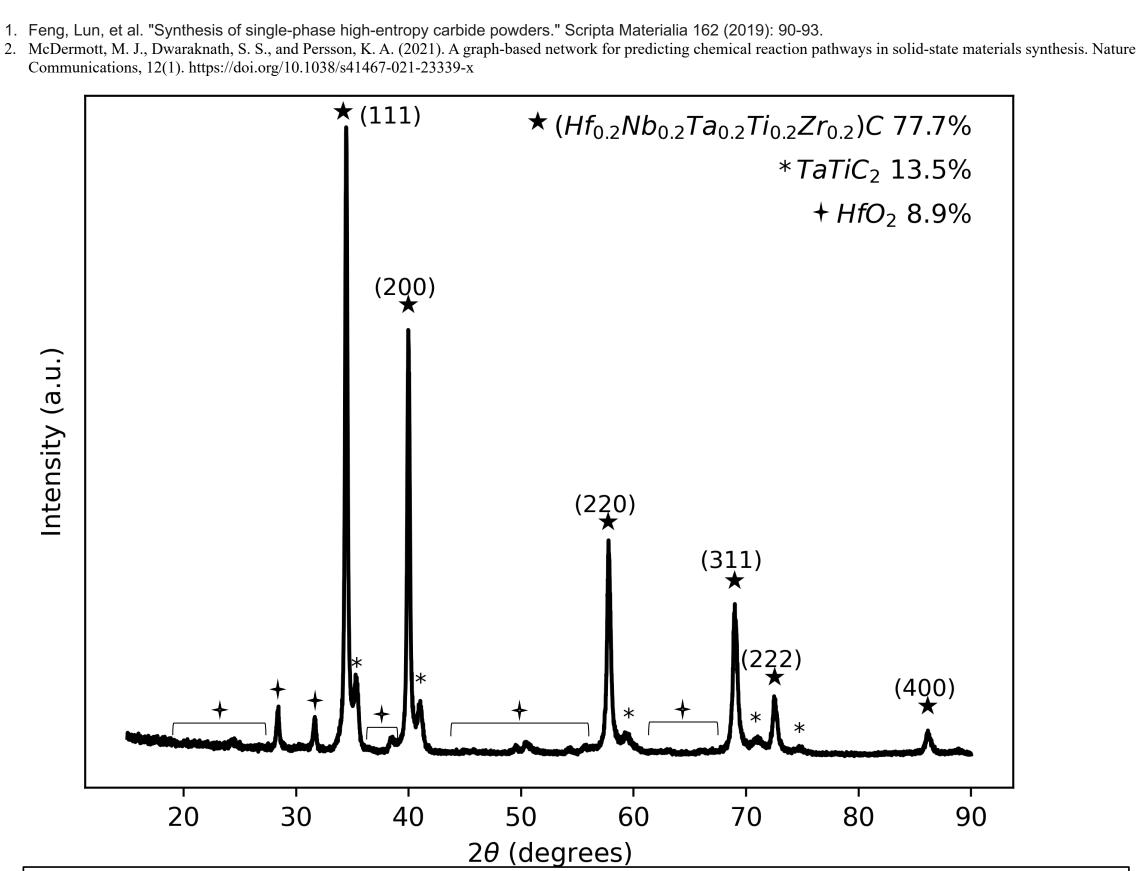
# **Progress on High-Entropy Carbides:**

#### Carbothermal Reduction (CTR) Reactions At 1600°C

 $HfO_{2(s)} + 3C_{(s)} \to HfC_{(s)} + 2CO_{(g)} (dG = 0.131 \text{ eV/atom})$  $ZrO_{2(s)} + 3C_{(s)} \rightarrow ZrC_{(s)} + 2CO_{(g)} (dG = -0.041 \text{ eV/atom})$  $TiO_{2(s)} + 3C_{(s)} \rightarrow TiC_{(s)} + 2CO_{(g)} (dG = -0.177 \text{ eV/atom})$  $1/2Ta_2O_{5(s)} + 7/2C_{(s)} \rightarrow TaC_{(s)} + 5/2CO_{(g)} (dG = -0.3 \text{ eV/atom})$  $1/2Nb_2O_{5(s)} + 7/2C_{(s)} \rightarrow NbC_{(s)} + 5/2CO_{(g)} (dG = -0.323 \text{ eV/atom})$ 

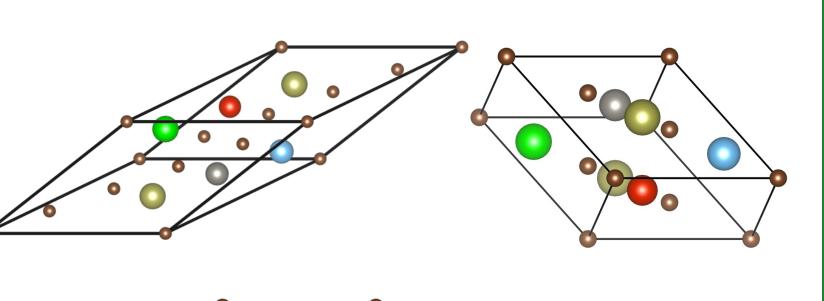
#### Solid Solution (SS) Formation At $2000^{o}$ C

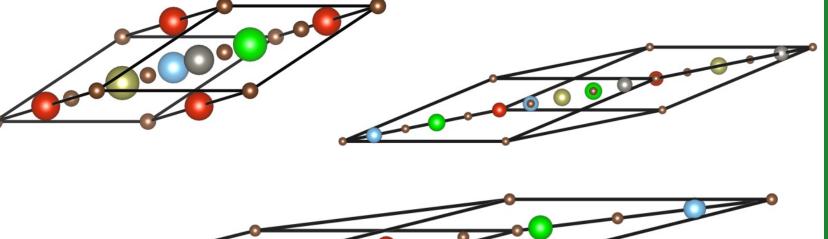
 $1/5(HfC_{(s)} + ZrC_{(s)} + TiC_{(s)} + TaC_{(s)} + NbC_{(s)}) \rightarrow (Hf_{0.2}Zr_{0.2}Ti_{0.2}Ta_{0.2}Nb_{0.2})C$ 

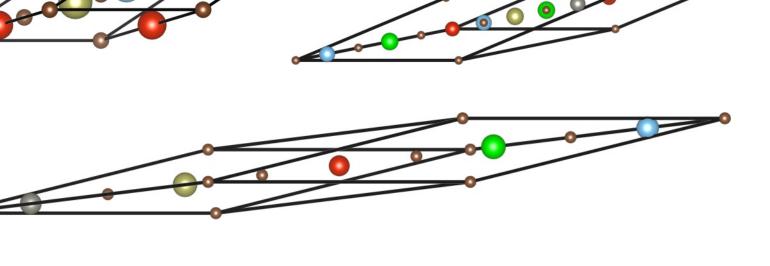


X-ray diffraction after MWPP of precursor metal-oxide powders (with graphite) to form HEC based on (Hf<sub>0.2</sub>Nb<sub>0.2</sub>Ta<sub>0.2</sub>Ti<sub>0.2</sub>Zr<sub>0.2</sub>)C Lattice parameter from literature<sup>1</sup>: 0.4524 nm

**Lattice parameter from the experiment:** 0.4517 nm

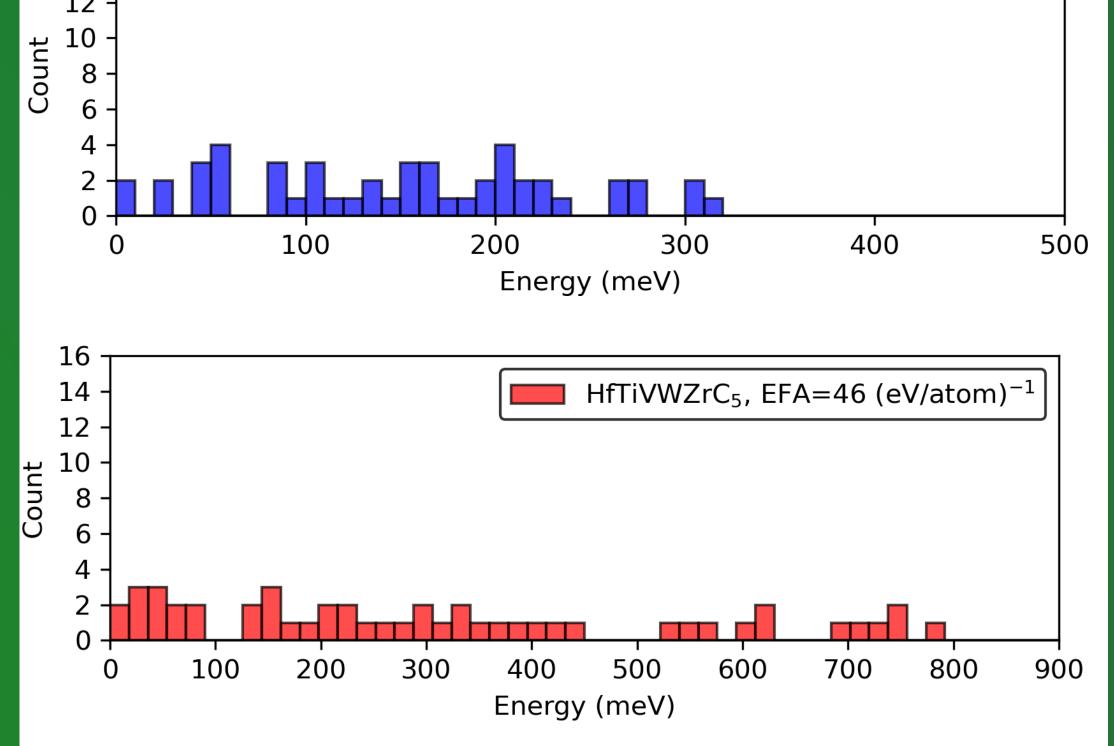








### **Computational Modeling of Entropy** Forming Ability (EFA):



$$EFA(N) \equiv \left\{ \sigma[\operatorname{spectrum}(H_i(N))]_{T=0} \right\}^{-1}$$

## Challenges:

- Resistance to increase in temperature in the range of 1700-2000° C
- Large fluctuations in the temperature
- Gibbs free energy for CTR of  $HfO_{2(s)}$  is positive
- Hf is limiting element for SS reaction

### Conclusion:

Predominately single-phase HEC can be made by MW plasma-annealing

### **Future Work:**

Investigate the role of excess graphite reducing agent as well as temperature and time to increase the propensity toward full conversion to the HEC single phase.

# Acknowledgement

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